# ZnON MIS Thin-Film Diodes

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*Abstract*— ZnON metal-insulator-semiconductor (MIS) diodes are fabricated and characterized. Although these devices display excellent rectification, their temperature-dependent current-voltage (I-V) characteristics are not explicable using analysis methodologies currently available in the literature. Therefore, we employ a simple curve fitting strategy in order to elucidate measured trends. It is found that the forward current trends are describable using three parameters, i.e., reverse saturation current, ideality factor, and series resistance, whereas the reverse current temperature dependence only requires one parameter, i.e., shunt resistance. All four of these model parameters are found to be strongly temperature dependent.

*Index Terms*—metal oxide semiconductor, zinc oxynitride, MIS diode, thin film device

### I. INTRODUCTION

CTIVE metal oxide semiconductor materials have been a Afocus in recent years due to their superior performance. Amongst other advantages, low cost, the possibility of low temperature deposition, and robust integration with other organic and inorganic compounds are key attributes. In this paper, we report on the characterization of diodes fabricated from an oxide semiconductor, with the ultimate objective of exploring the possibility of using such diodes in energy harvesting applications. One of the most extensively studied semiconductors in this group is zinc oxide (ZnO). Several ZnO diode studies have been reported [1]–[3]. In order to improve the performance of ZnO, cation and anion control strategies have been investigated and implemented. These include the formation of InGaZnO and InZnO for metal cation control and ZnON for anion control. ZnON is an attractive semiconductor material since it possesses a high mobility ( $\approx 100 \text{ cm}^2/\text{Vs}$ ), high illumination stability, and the ability to be deposited at room temperature [4]–[6].

This work is supported by Engineering and Physical Sciences Research Council (EPSRC) under Project EP/M013650/1.

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Moreover, it has been shown that depending on the degree of nitrogen replacement of oxygen, the properties of ZnON can vary between ZnO and  $Zn_2N_3$  [4]. However, since ZnON is a semiconductor with a mixture of ZnO,  $ZnO_xN_y$ , and  $Zn_3N_2$ , and since Zn possesses a higher reactivity with oxygen compared to nitrogen, ZnON has been associated with problems of stability and reproducibility, especially when exposed to air. To counter this problem, several solutions have been proposed, including argon plasma treatment [7] and heat treatment [8][9]. In this work, we investigate the implementation of zinc oxynitride (ZnON) in a metal-insulator-semiconductor (MIS) thin-film diode structure.

The main purpose of depositing a very thin insulating layer between the metal and semiconductor is to minimize reaction and diffusion between metal and semiconductor [10]. Furthermore, the thin insulating film has been demonstrated to reduce leakage current and defect trapped charge density [11][12], thereby improving the reliability and performance of the diode. Another advantage is the possibility of reducing the interface state density [13][14]. In this contribution, aluminium oxide (Al<sub>2</sub>O<sub>3</sub>) is used as the insulating layer since the film quality and the thickness can be precisely controlled via atomic layer deposition (ALD).

ZnON MIS diode current-voltage (I-V) characteristics are modelled using [15]

$$I = I_0 \left[ exp\left(\frac{q(V - IR_s)}{nk_BT}\right) - 1 \right] - \frac{V - IR_s}{R_{sh}} \qquad (1)$$

where  $I_0$  is reverse saturation current, q is electronic charge,  $R_s$  is series resistance, n is ideality factor,  $k_B$  is Boltzmann's constant, T is temperature, and  $R_{sh}$  is shunt resistance. Since the exponential term dominates when V is positive, the forward current is describable using three model parameters, i.e.,  $I_0$ , n, and  $R_s$ . Conversely, since  $I_0$  and  $IR_s$  are found to be negligibly small at negative voltage for the ZnON MIS diodes tested, the reverse current is determined by only a single model parameter, i.e.,  $R_{sh}$ . In order to accurately fit experimental temperature-dependent I-V curves using eq. (1), all four model parameters are assumed to depend on temperature.

Although many parameter estimation strategies have been proposed in the literature for the direct extraction of  $I_0$ , n,  $R_s$ , and  $R_{sh}$  [16]–[22], we have been unsuccessful when employing these methods to the assessment of our ZnON MIS diodes. In contrast, we can readily estimate  $I_0$ , n,  $R_s$  using the exponential portion of eq. (1) to accurately fit to an experimental I-V curve. Moreover, we find our estimates of these parameters to be almost unique since the low, intermediate, and high current portions of the forward current are primarily determined by  $I_0$ , n, and R<sub>s</sub>, respectively as shown in Fig. 1. From this, a fitting procedure is proposed as follows. First I<sub>0</sub> is estimated as the y-intercept of the low voltage portion of the curve. Next using this estimated value of I<sub>0</sub>, n is varied whilst maintaining R<sub>s</sub> as 0  $\Omega$  until a good fit of the low voltage portion of the curve is obtained (this region is found to be approximately linear when plotted on a semilog scale). Next, R<sub>s</sub> is increased until a good fit to the higher voltage portion of the curve is obtained. Finally, all 3 parameters are tweaked to produce an optimal fit. This fitting is accomplished using Mathematica.



Fig. 1. The effects of all three parameters on the J-V curve. Black lines depict the actual behavior and the red, green, and yellow lines show the effects of varying the (a) ideality factor n, (b) series resistance Rs, and (c) saturation current Io, respectively whilst maintaining identical values for the other model parameters.

#### II. FABRICATION AND EXPERIMENTAL SETUP

The structure of the fabricated diode is shown in Fig. 2. The 1 mm thick Corning substrate was cleaned by first rinsing it with deionized (DI) water and then ultrasonicating it in acetone and then isopropyl alcohol solution, each for 10 minutes. It was then blow dried with nitrogen gas and subsequently baked at 120 °C for 30 minutes to remove possible moisture. Molybdenum bottom contact (100 nm) was then sputtered at room temperature. The ZnON layer (50 nm) was deposited through the reactive magnetron sputtering method by using zinc target (99.99 % purity) together with argon, nitrogen, and oxygen gases (Ar/N<sub>2</sub>/O<sub>2</sub> = 4/180/1.5 SCCM) that flowed into the chamber at room temperature. The chamber pressure and RF power were maintained at 12.5 mTorr and 250 W, respectively. From the Hall measurements, the magnitude of the Hall mobility for the as-deposited ZnON film based on this formulation is 21.5 cm<sup>2</sup>/Vs. The Al<sub>2</sub>O<sub>3</sub> layer (2.5 nm) was then deposited through the atomic layer deposition (ALD) at 250 °C. Finally, a nickel top contact (100 nm) was sputtered at room temperature. The sample was then annealed at 250 °C in normal atmospheric condition for 2 hours. The annealing condition was optimised based on the improvements/degradations observed in the forward I-V characteristics after each annealing process. Shadow mask was used to pattern the semiconductor, insulator, and top contact layer. The active area of the diode is 0.66 mm<sup>2</sup>.

I-V and C-V measurements were carried out using Keithley 4200 Semiconductor Characterization System (4200-SCS) inside a LakeShore probe station and the temperature was controlled using Lakeshore 336 Temperature Controller. All measurements were carried out in the dark and the temperature was varied from 50 °C to 200 °C in 50 °C intervals (323 K to 473 K in 50 K intervals). All I-V measurements were done under a vacuum environment except for room temperature measurement (normal atmosphere). The voltage for the I-V characterization for the room temperature measurements and temperature dependent measurements was swept from -3 V to +3 V with 0.2 V intervals and from -1.5 V to 1.5 V with 0.05 V intervals, respectively. The C-V sweep was carried out at 1 MHz with an ac modulation voltage of 30 mV



Fig. 2. MIS diode structure and actual sample of the fabricated diode. All layers are sputter deposited except for the  $Al_2O_3$  which is by atomic layer deposition.

## III. RESULTS AND DISCUSSION

## A. Room Temperature I-V and C-V Measurements

A J-V plot for a ZnON MIS diode at room temperature is shown in Fig. 3. The diode shows typical rectifying behaviour (exponential increase of current in forward bias and weak dependence in reverse bias) with a threshold voltage of approximately 0.95 V and a leakage current density of 0.4  $\mu$ A/cm<sup>2</sup> at -3 V. Introduction of a thin interfacial layer between metal and semiconductor reduces the leakage current and likely increases the threshold voltage [12]. The rectification ratio of the diode is 9500 at ± 3 V. The forward and reverse breakdown voltage of the diode are 28 and -25 V, respectively. Generally, the diode possesses a strong rectifying property.

The forward current 3-parameter curve fitting method is implemented to obtain n = 2.55,  $I_0 = of 1.6 \times 10^{-10}$  A, and  $R_s = 3800 \Omega$ . The reverse current 1-parameter fitting yields  $R_{sh} = 1.3 G\Omega$ .

The n = 2.55 ideality factor is much larger than that expected for thermionic emission (n = 1) or generation-recombination (n = 2). This large value of n can be elucidated using induced gap state (IGS) theory [14], as follows. Modeling the ZnON MIS diode in non-equilibrium as a voltage divider consisting of an insulator capacitance density,  $C_I$ , in series with a parallel combination of the semiconductor and interface state capacitance densities leads to

$$n \approx 1 + \frac{C_{FB} + C_{SSmin}}{C_I} = 1 + \frac{3.6 \, F \, cm^2 + 1.7 \, F \, cm^2}{3.2 \, F \, cm^2}$$
$$= 2.67 \qquad (2)$$

where  $C_{FB}$  and  $C_{SSmin}$  are flatband and minimum interface state capacitance densities, which apply when the semiconductor is operating in depletion. This calculated value of n = 2.67 is in reasonable agreement with the measured value of n = 2.55. According to IGS theory, current in the ZnON MIS diode is thermionic, but n is large since much of the applied voltage drops across the insulator instead of the semiconductor.



Fig. 3. J-V and C-V curves for a ZnON MIS diode at room temperature. Inset (a) shows the  $1/C^2$ -V curve that is plotted in order to estimate the barrier height. Inset (b) shows an I-V plot in reverse bias, which is almost linear and is used to estimate the shunt resistance of the diode.

Parameter	Magnitude			
Ideality factor, n	2.55			
Series resistance, R <sub>s</sub>	3800 Ω			
Barrier height from I-V, $\Phi_{\rm B}$	0.83 eV			
Barrier height from C-V, $\Phi_{\rm B}$	0.86 eV			
Shunt resistance, R <sub>sh</sub>	1.3 GΩ			

The free electron carrier concentration is determined from the slope of the  $1/C^2$ -V plot as shown in Fig. 3 and inset (a).The barrier height is calculated from the built-in potential estimated by extrapolating the  $1/C^2$ -V plot [23]. From this, the calculated free electron carrier concentration and barrier height are  $2.2 \times 10^{18}$  cm<sup>-3</sup> and 0.86 eV, respectively. An alternative way to estimate the barrier height,  $\Phi_B$ , is from I-V data using [23]

$$I_0 = AA^{**}T^2 \exp\left(-\frac{q\Phi_b}{k_BT}\right) \qquad (3)$$

where A is diode area and  $A^{**}$  is the effective Richardson constant [24]

$$A^{**} = \frac{4\pi q m^* k^2}{h^3} \qquad (4)$$

where m\* is effective mass and h is Planck's constant. Assuming m\* = 0.19 m<sub>0</sub> for ZnON [6][25] and using I<sub>0</sub>=1.6 x  $10^{-10}$  A leads to  $\Phi_B = 0.83$  eV. Thus, our C-V and I-V estimates for  $\Phi_B$  are quite similar. Parameters extracted from room temperature I-V and C-V measurements are summarized in Table 1. Table 4 presented at the end of this contribution provides overall comparison of the parameters extracted at room temperature for the diode fabricated in this contribution and other reported metal oxide diodes work.

#### B. Temperature Dependent I-V Measurements

Fig. 4 shows J-V curves in forward bias for a ZnON MIS diode at four temperatures. J-V characteristics depend strongly on temperature, increasing with increasing temperature. The 3-parameter fit to the measured J-V curves is quite good. Extracted fitting parameters for all curves are collected in Table 2. From Table 2, it can be observed that all four model parameters vary strongly with temperature.



Fig. 4. Measured (indicated as points) and fits (smooth curves) of the forward J-V curves for a ZnON MIS diode at multiple temperatures.

Table 2. Model parameters used to fit measured temperature-dependent I-V curves for a ZnON MIS diode.

Temp. (K)	<b>I</b> <sub>0</sub> ( <b>A</b> )	n	$R_{S}(\Omega)$	$\Phi_{\rm B} \left( {\rm eV} \right)$
323	6.15 x 10 <sup>-10</sup>	2.43	7700	0.86
373	1.68 x 10 <sup>-9</sup>	1.90	3550	0.97
423	1.12 x 10 <sup>-8</sup>	1.72	1050	1.04
473	6.56 x 10 <sup>-8</sup>	1.65	420	1.10



Fig. 5. Measured (indicated as points) and fits (smooth curves) of the reverse biased curves for a ZnON MIS diode at multiple temperatures.

measured reverse bias I-V curves for a ZnON MIS diode. Temp. (K)  $R_{sh}(G\Omega)$ 323 0.50373 0.33 423 0.05 473 0.01 8x10<sup>-6</sup> 1.1x10 2.4 - Фı 6x10-6 1 1x10 22 Фв (eV) 1.0x10 4x10 2.0 9.5x10 0 2×10-8 1.8 9.0x10 C 8.5x10 16 300 400 450 500 300 350 400 450 500 350 Temperature (K) Temperature (K) (b) (a) 6x10 8000 R = -48.58T + 22502.3  $R_{ab} = -3.5E6T + 1.6E9$  $R^2 = 0.85$  $R^2 = 0.91$ 4x10<sup>6</sup> 6000 (C) (C) کر 2x10<sup>6</sup> 4000 ഹ് 2000 0 450 300 300 350 400 500 350 400 450 500 Temperature (K) Temperature (K) (d) (c)

Table 3. Shunt resistance,  $R_{sh}$  as a function of temperature as estimated from

Fig. 6. Temperature trends for model parameters used to obtain J-V fits to measured data for a ZnON MIS diode. (a)  $I_0$  and  $\Phi_B$ , (b) n, (c) R<sub>s</sub>, and (d) R<sub>sh</sub>.

Next, the reverse bias I-V curves as the function of temperature for a ZnON MIS diode are shown in Fig. 5. Using eq. (1), a linear fit of the data is used to estimate the shunt resistance. These values of R<sub>sh</sub> are compiled in Table 3.

All of the five extracted parameters; saturation current  $I_0$ , barrier height  $\Phi_B$ , ideality factor n, series resistance  $R_s$ , and shunt resistance R<sub>sh</sub> are strongly temperature dependent; temperature trends for each of these parameters are plotted in Fig. 6. I<sub>0</sub> increases with increasing temperature, as does  $\Phi_{\rm B}$ since they trend together according to eq. (3). n decreases with increasing temperature, suggesting that thermionic emission contributes more to the forward current at higher temperatures. Both R<sub>s</sub> and R<sub>sh</sub> decrease with increasing temperature. This is ascribed to an increase in the free electron carrier concentration with increasing temperature. R<sub>sh</sub> is suspected to arise from the existence of a peripheral leakage path at the edges of the diode. Thus, the decrease of R<sub>sh</sub> with increasing temperature is also ascribed to an increase in the electron free carrier concentration with increasing temperature.

Evidence that the physical mechanism responsible for the temperature dependence of Rs and Rsh is identical is provided by the observation that their fractional decrease in resistance with increasing temperature are quite similar, i.e.,  $R_s$  (T) /  $R_s$  $(323 \text{ K}) = 1, 0.46, 0.14, \text{ and } 0.05; R_{\text{sh}}(\text{T}) / R_{\text{sh}}(323 \text{ K}) = 1, 0.66,$ 0.1, and 0.02 for T = 323, 373, 423, and 473 K. The free electron carrier concentration is estimated for each temperature by assuming a linear relation of R<sub>s</sub> with temperature and using the free electron carrier concentration at room temperature as obtained from C-V measurements in the previous section

$$R_s(T) = \frac{\rho(T)L}{A} \approx \frac{L}{q\mu_n n_c(T)A} \propto \frac{1}{n_c(T)}$$
(5)

where  $\rho$  is resistivity, L is length,  $\mu_n$  is electron mobility, and  $n_c$ is free electron carrier concentration. By assuming that other parameters apart from R<sub>s</sub> and n<sub>c</sub> are independent of temperature, the estimated temperature-dependent free electron carrier concentration obtained is shown in Fig. 7. The regression fit to the data reveals a characteristic temperature of 50 K, corresponding to a free electron carrier concentration enhancement energy of 4 meV.



Fig. 7. Temperature-dependent free electron carrier concentration estimated from the temperature-dependent Rs data.

## IV. CONCLUSION

In this contribution, ZnON MIS diodes are fabricated and characterized. These devices are found to exhibit excellent rectification. A fitting method is employed to describe the strongly temperature dependent forward and reverse current characteristics of the diode. Measured forward current characteristics are accurately fit using three model parameters, i.e., saturation current, ideality factor, and series resistance while measured reverse current characteristics are fit using only one parameter, i.e., the shunt resistance.

Ref.	Diode structure & material	Thickness (nm)	Ideality factor	Barrier height (eV)	Threshold voltage (V)	Carrier concentration (cm <sup>-3</sup> )	Series resistance (Ω)	Rectification ratio	Breakdown voltage (V)
[26] 2019	TiN/a-IGZO/Cu (MS)	110	1.3	0.76	0.6	5.7 x 10 <sup>17</sup>	-	$3 \ge 10^{6} at \pm 1$ V	-
[27] 2018	p-Si/n-ZnO:Al (pn)	-	1.96	0.70	-	0.9 x 10 <sup>11</sup>	6340	-	-
[28] 2018	Pd/IGZO/Ti/Au (MS)	180	1.09	0.80	-	-	-	$1.3 \times 10^{6} \text{ at } \pm 1 \text{ V}$	-7.5
[29] 2018	Au/SnO <sub>2</sub> /p-Si (MS)	53.5 (Au and $SnO_2$ only)	1.11 – 1.40	0.33 – 0.63	-	1 x 10 <sup>20</sup>	-	-	-
[30] 2018	Au/AZO (MS)	150 (AZO only)	10	0.39	-	-	110	-	-
[31] 2017	Pt/IGZO/Al (MS)	100 (Pt and IGZO only)	1.36	0.92	0.6	4.05 x 10 <sup>18</sup>	507	$\begin{array}{c} 2.4 \ x \ 10^6 \ at \ \pm \\ 1 \ V \end{array}$	-
[32] 2017	Pd/IGZO (MS)	230	1.09	0.88	0.5	9.9 x 10 <sup>13</sup>	-	$7.2 \text{ x } 10^7 \text{ at } \pm 1 \text{ V}$	-
[33] 2017	PEDOT-PSS/a-IGZO (MS)	175	1.57	0.77	0.4	-	828	1.8 x 10 <sup>5</sup> at ± 1 V	-
[34] 2017	n-IZO/p-CuO (pn)	165	10	1.16	1.4	-	-	$10^4$ at $\pm 2$ V	-
[35] 2017	Pd/MgZnO/Al (MS)	-	1.87	0.62	1.9	-	-	59.9 at $\pm$ 5 V	-
[36] 2017	Pd/IGZO/Ti (MS)	230	1.15	0.73	-	2 x 10 <sup>14</sup>	33	$6.8 \text{ x } 10^4 \text{ at } \pm 1 \text{ V}$	-6
[37] 2016	Al/n-ZnO/p-Si (pn)	110 (Al and ZnO)	2.36	0.41	-	-	-	-	-
[38] 2016	Pd/IGZO/Ti (MS)	230	1.14	0.73	-	-	15	$1.3 \times 10^5 \text{ at } \pm 1 \text{ V}$	-6
[39] 2016	Pd/Zn-Sn-O/Mo (MS)	-	1.94	0.47	-	1.13 x 10 <sup>17</sup>	20	$10^2$ at $\pm 1$ V	-
[40] 2016	Al/ZnO/Au (MS)	10 (ZnO only)	1.9	0.79	-	1.4 x 10 <sup>18</sup>	28000	$10^{6} \text{ at } \pm 2.5 \\ V$	-
[41] 2016	p-NiO/ $\beta$ -Ga <sub>2</sub> O <sub>3</sub> (pn)	-	2	-	1.4	-	95	$10^8  at \pm 3 \; V$	-46
[42] 2016	Pd/IGZO/Ti/Au (MS)	196	1.23	1.02	-	1.5 x 10 <sup>13</sup>	284	$8.5 \times 10^7 \text{ at } \pm 2 \text{ V}$	-
[43] 2014	Pd/IGZO/Mo (MS)	230	1.2	0.83	-	-	250	$10^{10}$ at $\pm2$ V	-
[44] 2014	ITO/n-IGZO/p-Cu <sub>2</sub> O/Pt (pn)	340 (IGZO and Cu <sub>2</sub> O)	1.4	-	0.4	-	-	$3.4  ext{ x } 10^4  ext{ at } \pm 1.2  ext{ V}$	-
[45] 2013	Au/Ni/ZnO/ITO (MS)	1300	1.2	0.90		1.5 x 10 <sup>15</sup>	-	$3.3 \times 10^7 \text{ at } \pm 2 \text{ V}$	-
[46] 2012	Ag/MgZnO/ZnO/MgZnO/Ti/Au (MS)	280	2.37	0.73	0.7	-	1371	$10^3$ at $\pm 1$ V	-
[47] 2010	Al/ZnO/p-Si/In (pn)	70 (ZnO)	2.7	-	1.5	-	-	$3.3 \times 10^3 \text{ at } \pm 3 \text{ V}$	-
[48] 2010	Au/Ni/ZnO/ZnO-ITO/Al (MS)	2500	1.23	0.82	1.4	-	-	$1.8 \times 10^4 \text{ at } \pm 2 \text{ V}$	-
This	Ni/Al <sub>2</sub> O <sub>3</sub> /ZnON/Mo (MIS)	252.5	2.55	0.83	0.9	2.2 x 10 <sup>18</sup>	3800	9.5 x $10^3$ at ±	-19

Table 4: Comparison of the extracted parameters at room temperature between the diode fabricated in this contribution and other recent reported diode work based on metal oxide semiconductors.

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